

Some Trends of Synchrotron Radiation X-ray Imaging

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Outline

- **SR X-ray imaging**
 - 1 – where from do we come?
 - 2 – where are we?
coherent beams \Rightarrow new science
new aspects of microtomography
 - 3 – where should we go?
near future perspectives

- 1 – **SR X-ray imaging: where from do we come?**
- 2 – SR X-ray imaging: where are we?
 - # coherent beams \Rightarrow new science
 - “poled” ferroelectric domains
 - porosity and growth of quasicrystals
 - # new aspects of microtomography
- 3 – where should we go?
 - # near future perspectives

X-ray Imaging in the « Red book » of the ESRF

No actual imaging in the first phase of BLs construction

microfocus beamline \Rightarrow microfluorescence or
microdiffraction maps

Second phase:

- \Rightarrow Medical imaging
- \Rightarrow X-ray topography
- \Rightarrow microscopy and microanalysis

Microtomography

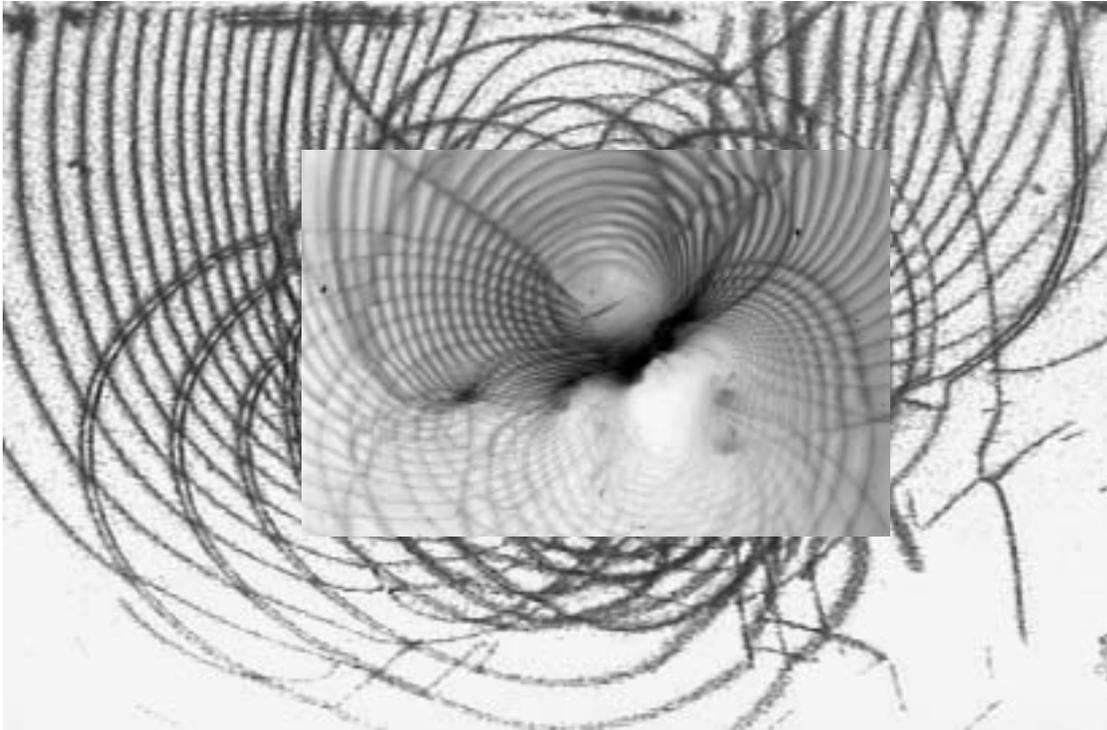
not foreseen in the « Red Book »

Reason

computer power

cluster of processors, 32 Mbyte of memory, 3 Gbytes of mass-storage

Diffraction imaging (“topography”) Frank-Read dislocation sources in SiC



White beam topograph showing dislocations originating from several Frank-Read sources, not actually visible (distorted region)

« Weak beam »
monochromatic topograph, which shows the location of the pinning centers (Frank-Read source)

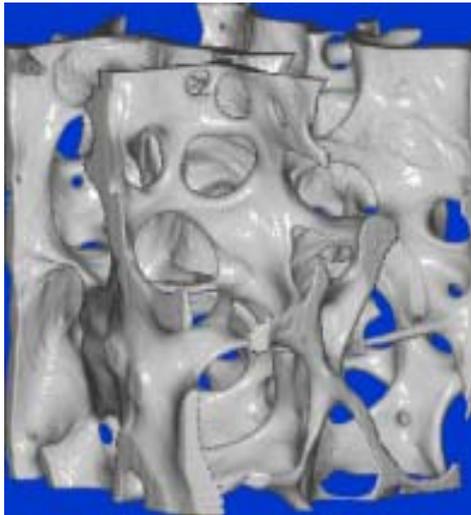
Eetu PRIEUR, PhD Thesis, Helsinki 1997

Microtomography

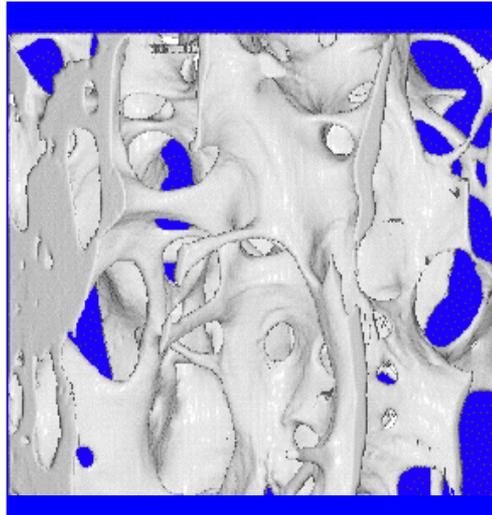
Osteoporosis disease

Human Vertebra Samples

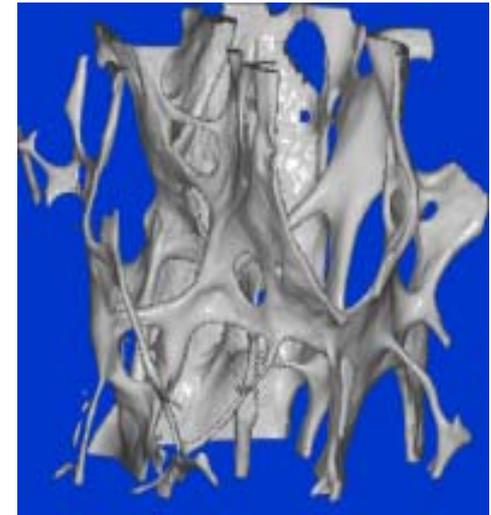
1mm



33 years



55 years



63 years

Images $(512)^3$, voxel size = $6.7 \mu\text{m}$

M. Salomé, F. Peyrin et al., *Medical Physics* **26** (1999) 2194-2204

•1 – SR X-ray imaging: where from do we come?

•2 – **SR X-ray imaging: where are we?**

coherent beams \Rightarrow new science

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X-ray imaging at the ESRF

About 15% of the scientific activity at the ESRF

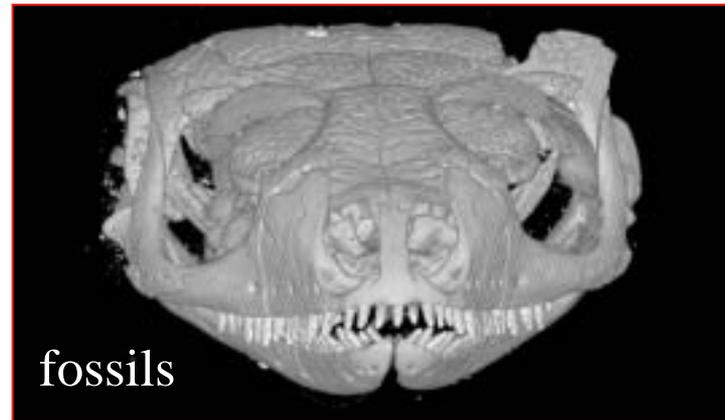
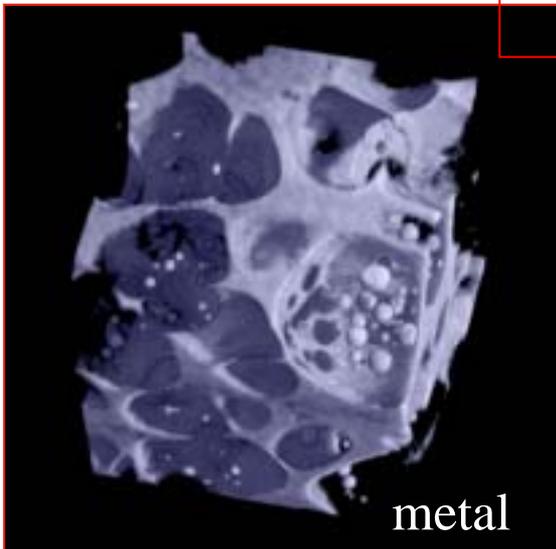
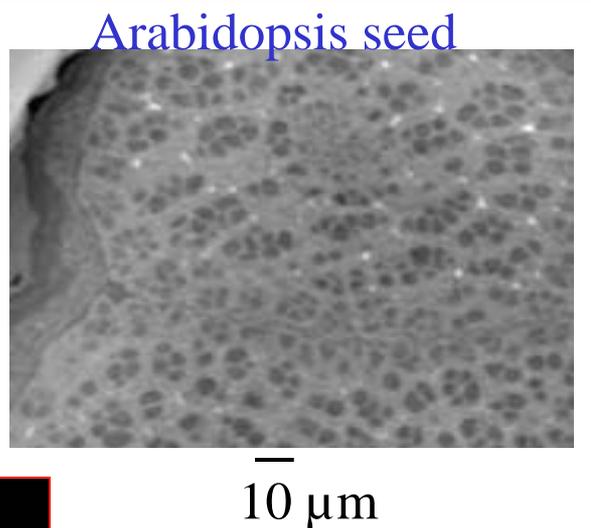
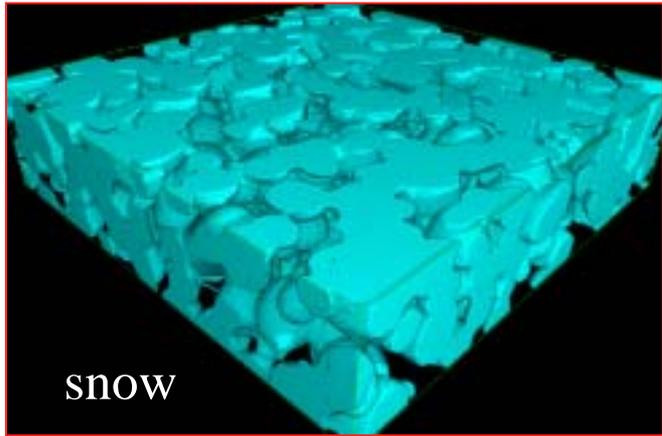
KEY WORDS: Medical imaging/ therapy, Use of Coherence, Microtomography and its applications, Combination of techniques, Microanalysis, Diffraction imaging

Increasing demand on microtomography and
microanalysis



enlargement of the number of scientific communities
concerned by these techniques

Microtomography: a multidisciplinary technique



Beamlines in the X-ray Imaging Group

BM05

6-100 keV
optics & imaging
(microtomography)

ID17

10-60 keV
medical imaging
& beam therapy

ID21(+ IR end station)

2-7 keV
imaging & microanalysis

ID19

6-60 keV
transmission & diffraction
imaging

ID22+ID18F

6-50 keV
imaging & microanalysis



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X-ray transmission imaging

Interaction of X-rays with matter: complex refractive index

$$n = 1 - \delta + i\beta$$

Absorption part expressed by $\beta \leftrightarrow \mu = 4\pi \beta / \lambda$

λ wavelength, μ linear absorption coefficient

Phase variation $\Delta\phi$ related to δ

$$\Delta\phi = 2\pi \delta t / \lambda$$

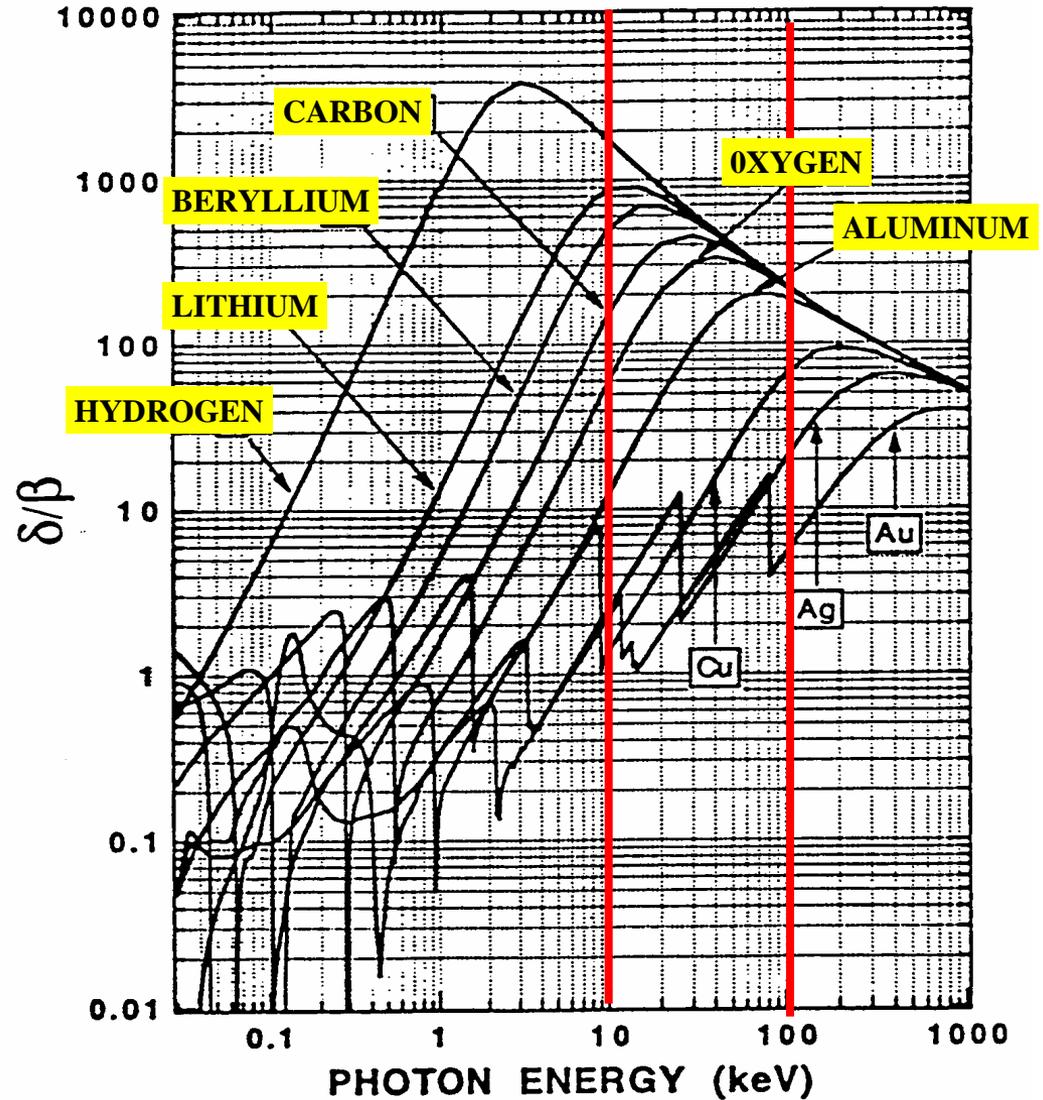
Index of refraction

$$n = 1 - \delta + i\beta$$

Ratio **phase** and **amplitude** (δ/β) as a function of the X-ray energy

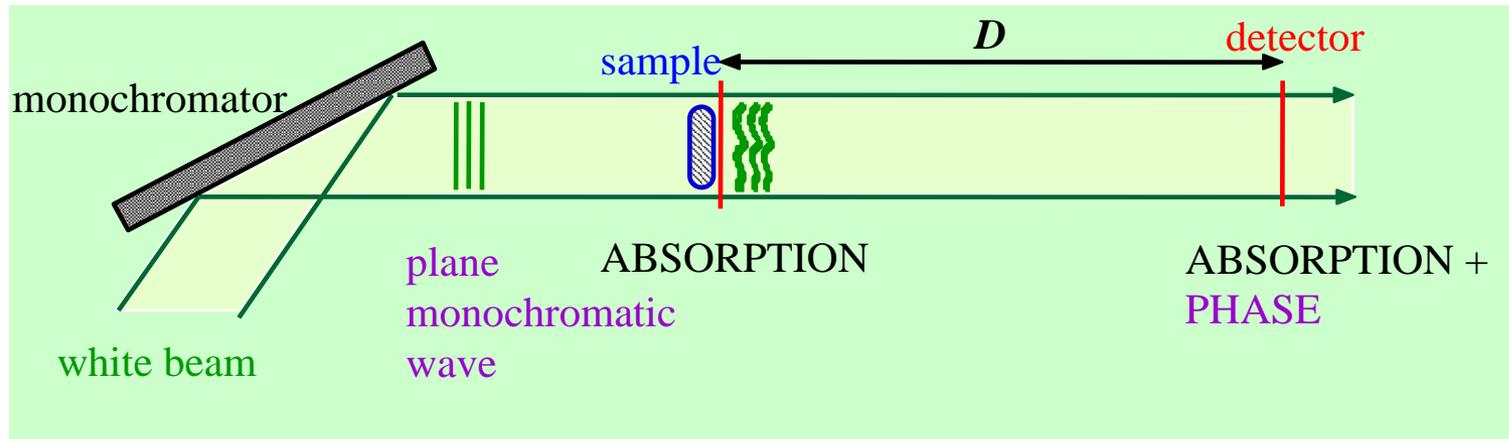
bigger than 10 in the hard X-ray range

Either reduced dose or enhanced sensitivity



coherent beam  Phase contrast imaging

the « propagation » technique



Two regimes: edge enhancement (only first fringes visible)
holograms

combined with high spatial resolution ($0.5\mu\text{m}$)
microtomography

Example: Damage in Composites

Scientific Case:

Quantification of *damage* in the *bulk* of composite materials
Plastic behaviour as a function of the hardness of the matrix

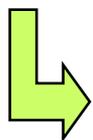
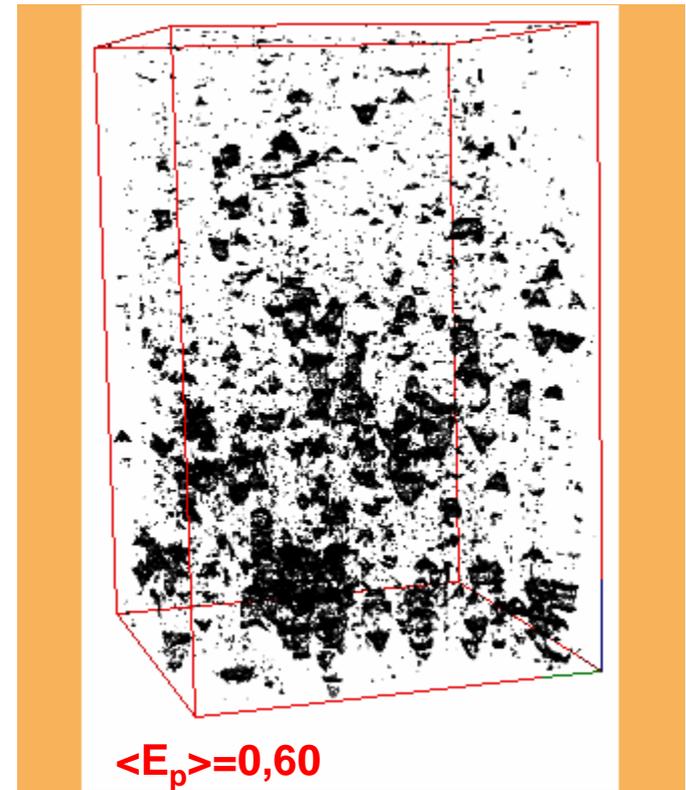
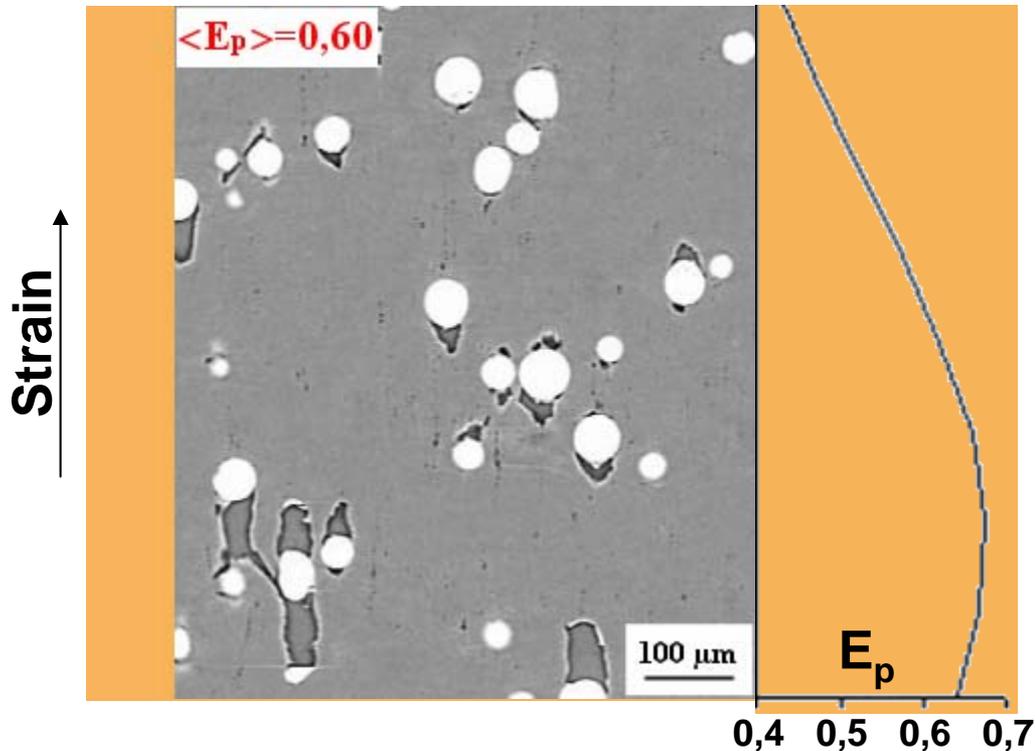
Authors:

L. Babout, W. Ludwig, E. Maire, J.Y. Buffière, *INSA, Lyon*
P. Cloetens, E. Boller, *ESRF, Grenoble*

Damage in Composites

L. Babout, W. Ludwig, E Maire, J. Y. Buffière
NIM B 200, 303 (2003)

Example: soft matrix
Al + 4%ZrO₂SiO₂



Preferred damage initiation: Matrix-particle *decohesion*

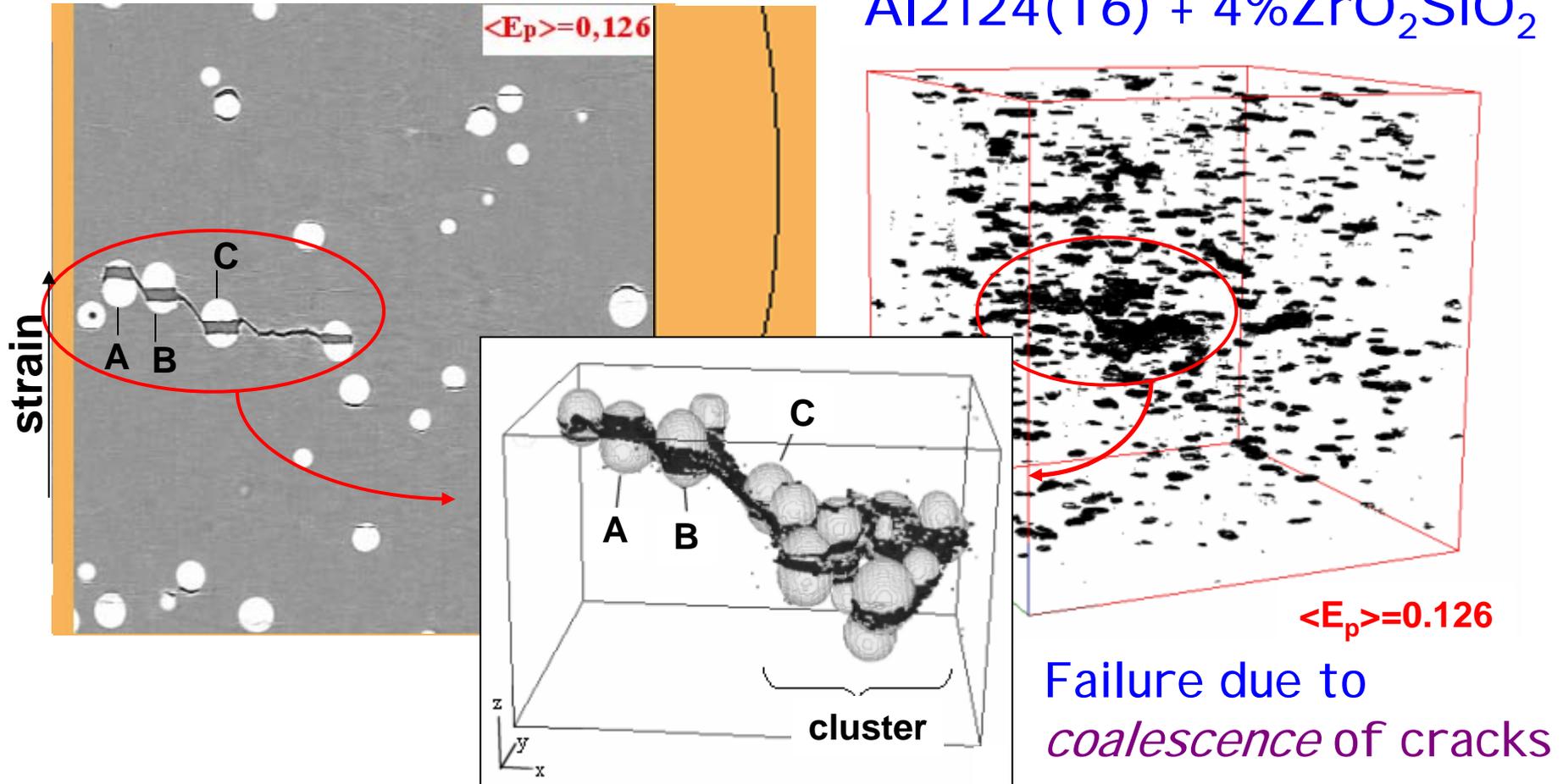
Conditions : in situ, E=30 keV abs/phase, 1 μm

Damage in Composites

L. Babout, W. Ludwig, E Maire, J. Y. Buffière

NIM B 200, 303 (2003)

Example: hard matrix
Al2124(T6) + 4%ZrO₂SiO₂



Preferred damage initiation: **particle rupture**

Conditions : in situ, E=30 keV abs/phase, 1 μ m

Phase imaging

propagation technique

What is measured $\Rightarrow \nabla^2\phi(y,z)$, phase ϕ second derivative

Phase retrieval \Rightarrow integral (measurements at several distances)

holo-tomography

- 1) phase retrieval with 3-4 distances $\phi_s | | |$
- 2) repeated for 700 views 

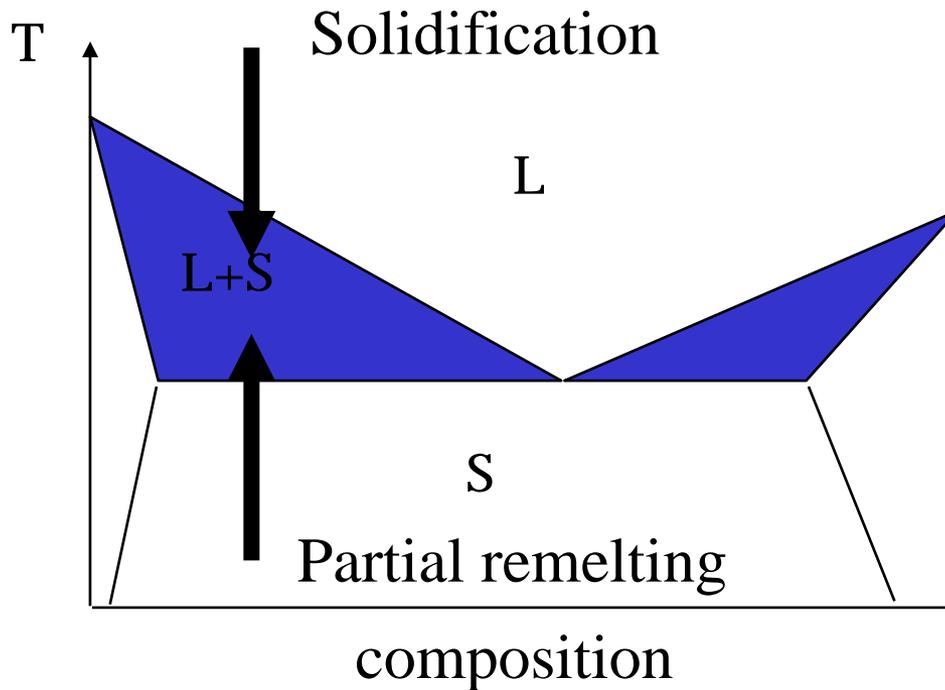
Experimental result \Rightarrow «local» $\phi(u,v)$

It can be shown that $\Delta \phi$ with respect to a propagation in vacuum is proportional to the local electronic density ρ_e $\phi \Rightarrow \rho_e$

Holotomography of *semi-solids*

P. Cloetens (ESRF), L. Salvo (GPM2, Grenoble)

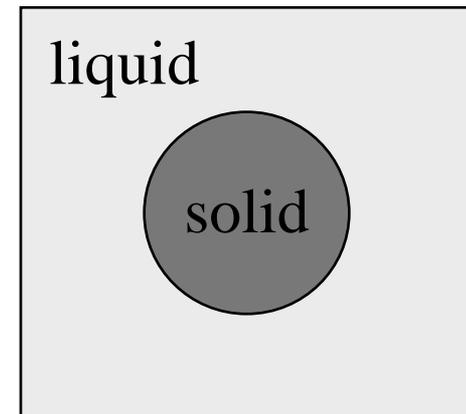
Rheology of aluminium alloys in the
semi-solid state Al – Al/Si system



Microstructure

solid fraction
morphology
connectivity

Classical Models



Holotomography on an Al - Al/Si system

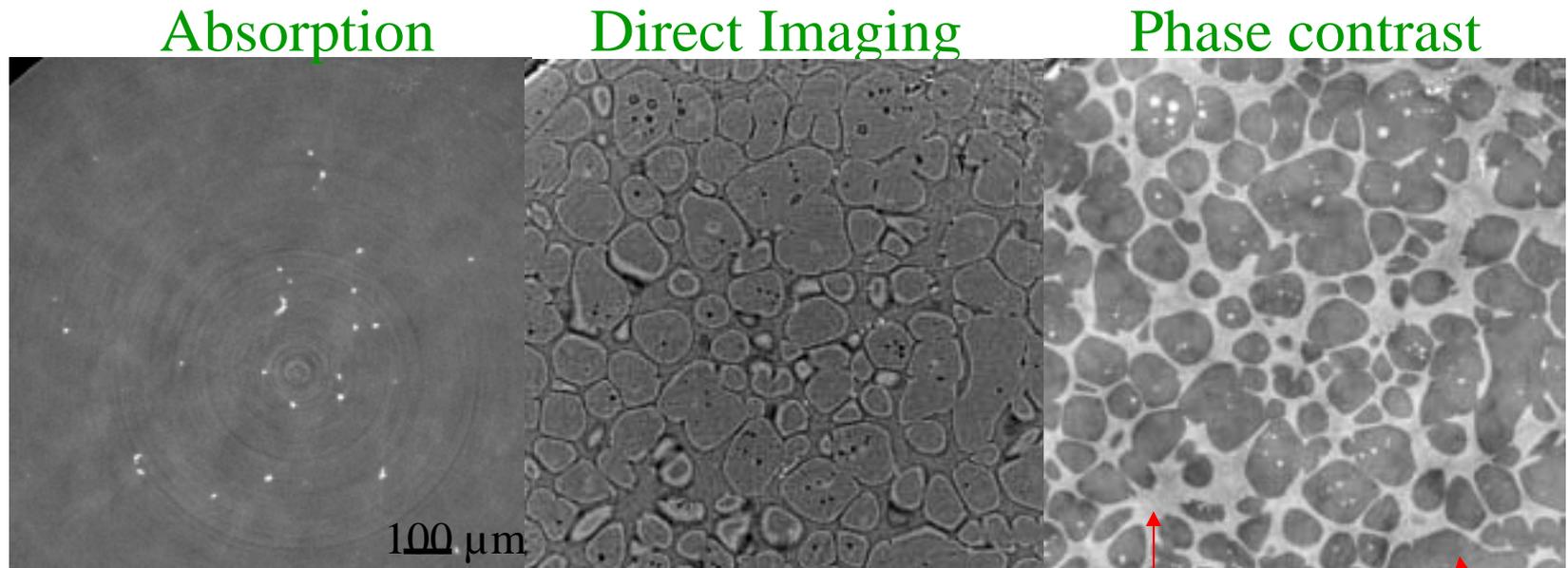
Rheology of aluminium alloys in the semisolid state

4 distances: absorption + 0.2 m, 0.5 m and 0.9 m

800 angular positions

multilayer as monochromator: total time \approx 40 minutes

E = 18 keV



β -map

edge enhancement

$D = 0.6$ m

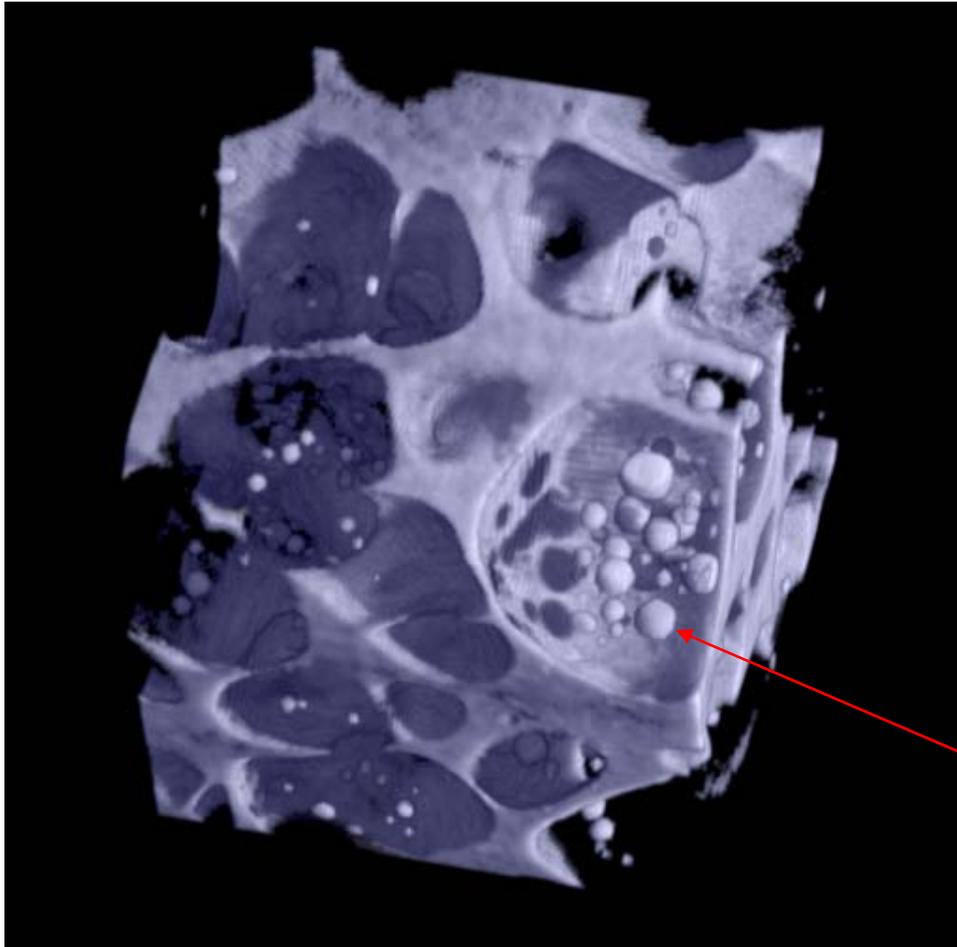
Al/Si

δ -map



Holotomography of *semi-solids*

Volume Rendering (solid transparent)



Connectivity:

Liquid phase:

total

+ trapped liquid

Solid phase:

very strong

trapped liquid

L. Salvo (GPM2, Grenoble) P. Cloetens (ESRF)

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PERIODIC FERROELECTRIC DOMAINS

P_S can be reversed by applying an electric field $> E_{\text{COERCIVE FIELD}}$

periodic electrodes defined on z+ face
high voltage pulses applied

⇒ Periodic ferroelectric domains

Diffraction by these domains:

domain I: F_{hkl}

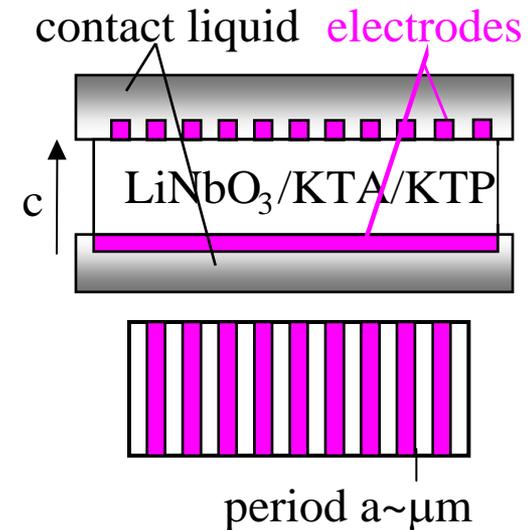
domain II: F_{-h-k-l}

$$|F_{hkl}| \sim |F_{-h-k-l}|$$

$$\text{but } \varphi_I \neq \varphi_{II}$$

⇒ phase shift $\Delta\varphi$

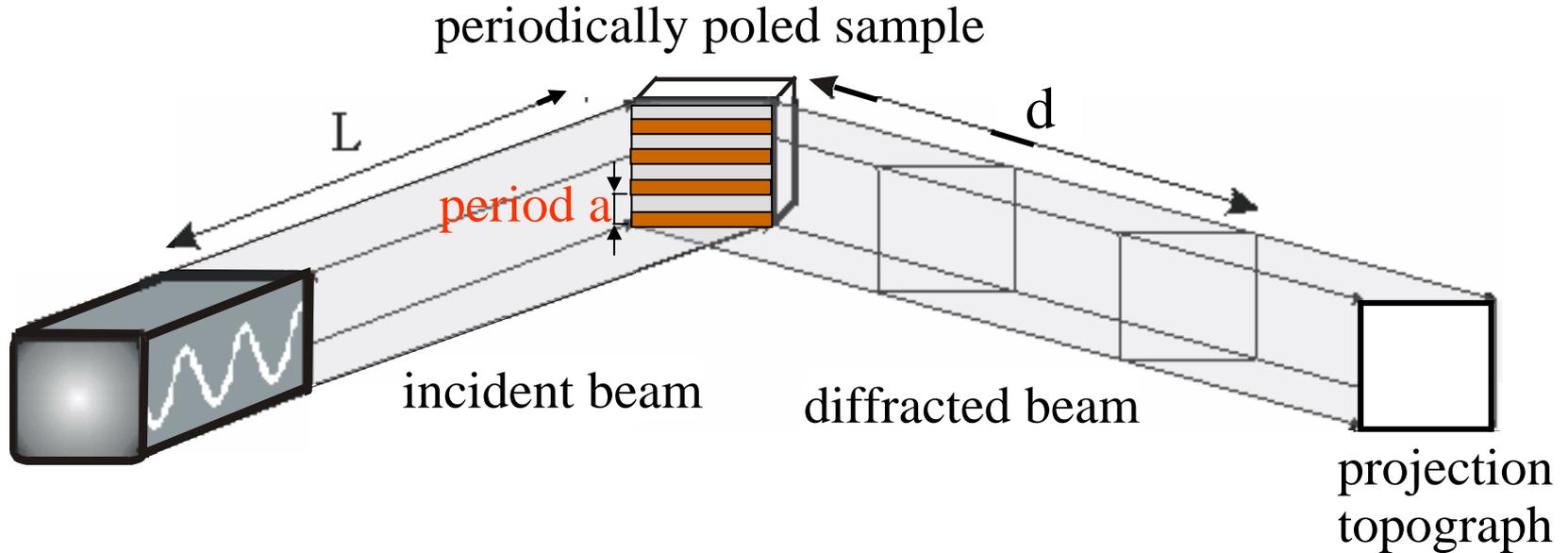
$$= \varphi_I - \varphi_{II}$$



BRAGG AND FRESNEL DIFFRACTION USED SIMULTANEOUSLY

$$l_c > a$$

⇒ visualisation of the inverted ferroelectric domains



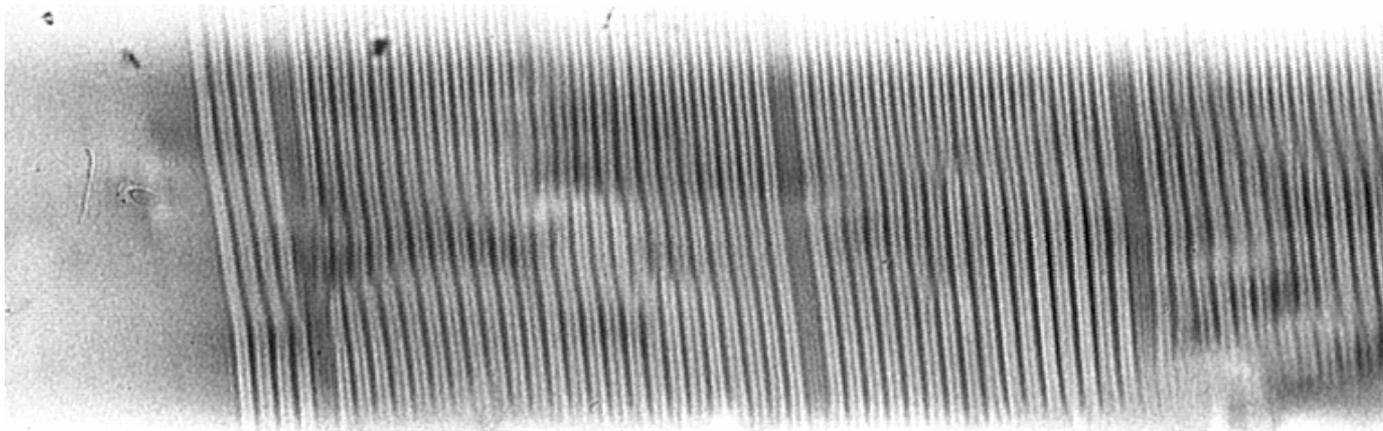
If wave diffracted by the adjacent domains exhibit a phase shift $\Delta\phi$

→ possibility to retrieve it via free space propagation

Visualisation of the domains in the bulk

Section topograph, transmission mode, where Bragg and Fresnel diffraction are combined to reveal the domains

KTP, period of poling $9\mu\text{m}$, Distance crystal-detector = 40 cm



Petra PERNOT, Pamela THOMAS, et al

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Quasicrystals

Quasicrystals (QCs) are a new class of solids, discovered in 1984 (Shechtman et al.), exhibiting:

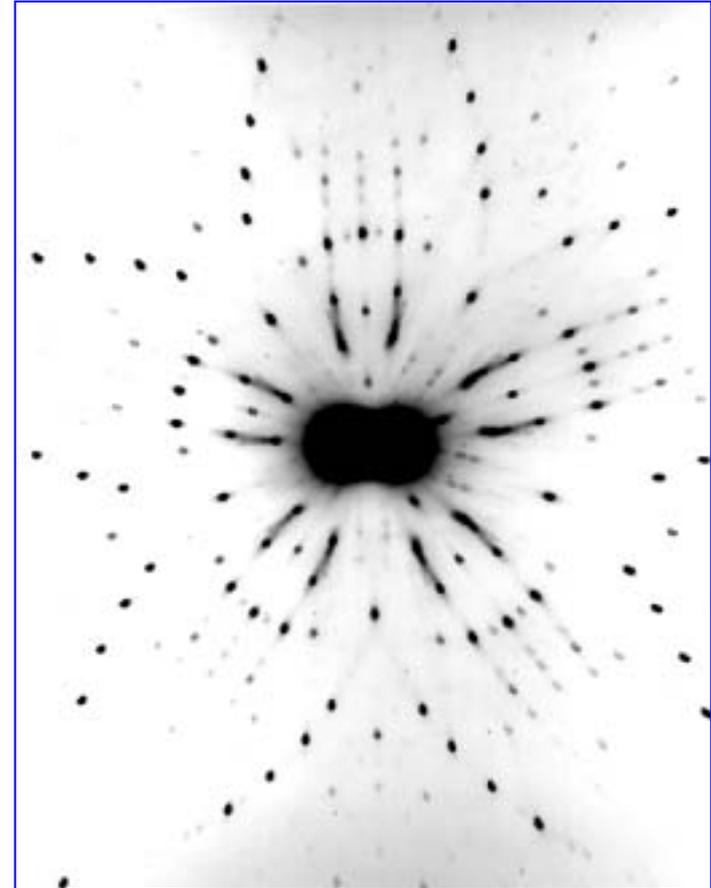
→ a long range *quasiperiodic* translational order

→ Orientational symmetries forbidden for crystals

(fivefold for instance)

→ sharp diffraction patterns

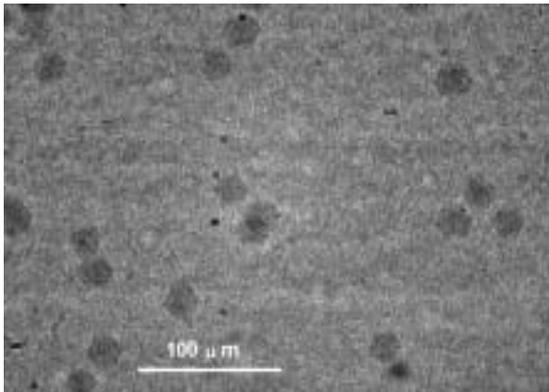
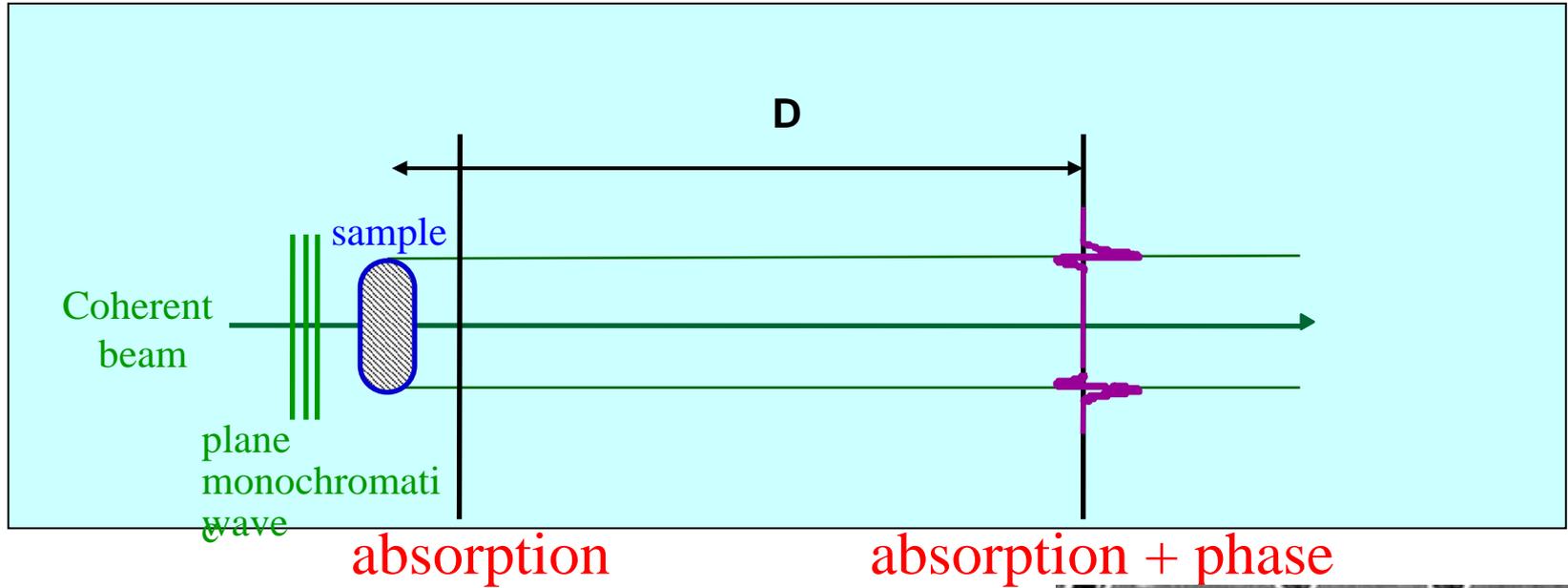
Laue image of an AlPdMn quasicrystal showing five-fold symmetry



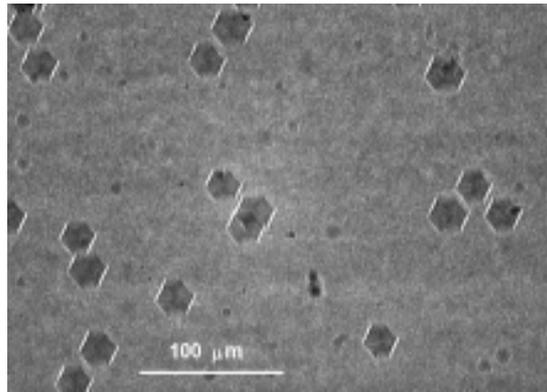
(J. Gastaldi)

- The complex structure of quasicrystals results in peculiar physical properties (e.g. high resistivity).
- Defects can influence these properties.
- One may detect precipitates, (dislocations), pores,
- Pores are defects commonly observed in quasicrystals.
⇒ origin and formation of porosity ?

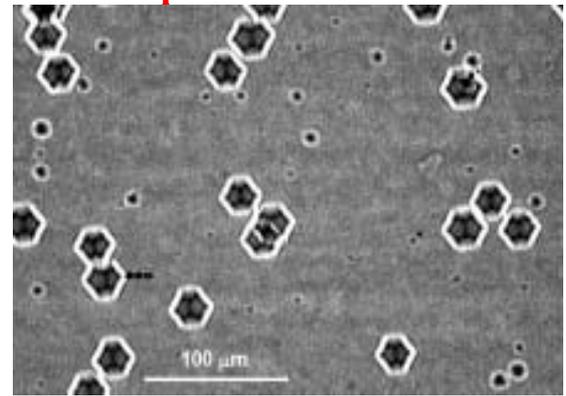
Phase Sensitive Radiography



$D = 1.3\text{ cm}$



$D = 10\text{ cm}$

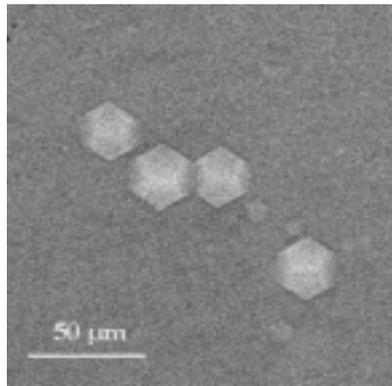


$D = 50\text{ cm}$

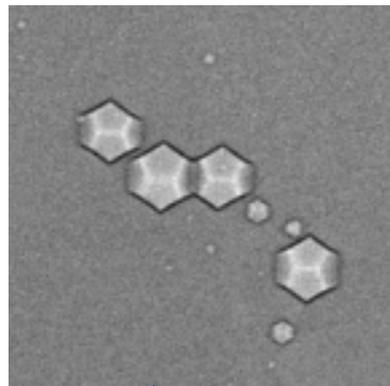
$\lambda = 0.35\text{ \AA}$

Determination of the real pore size

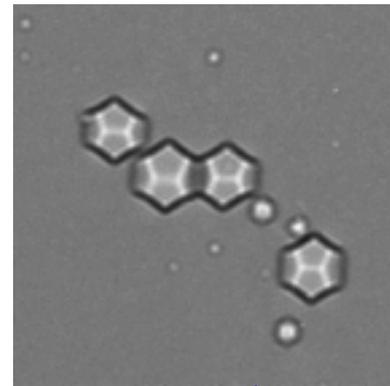
2-D phase reconstruction from recorded images at different distances



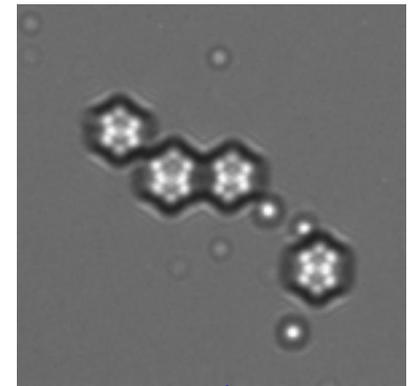
D=1cm



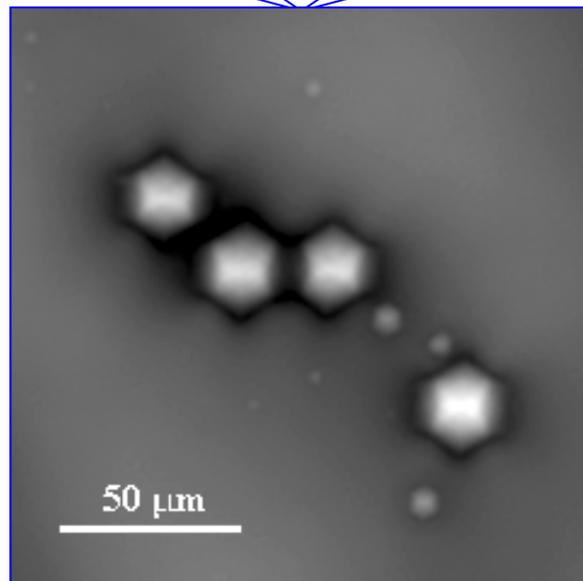
5cm



20cm

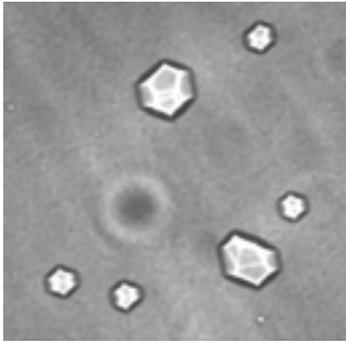


50cm

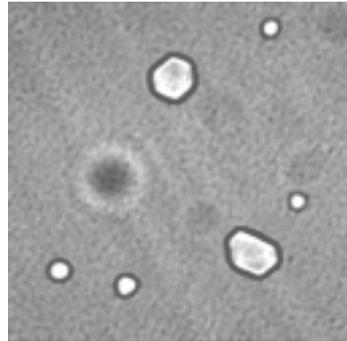


2-D phase reconstruction

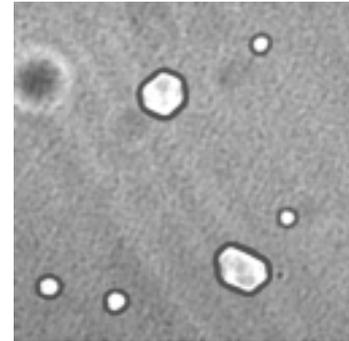
1st Annealing



a) RT ↗

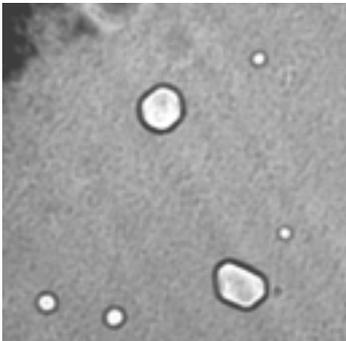


b) 800°C 3h25

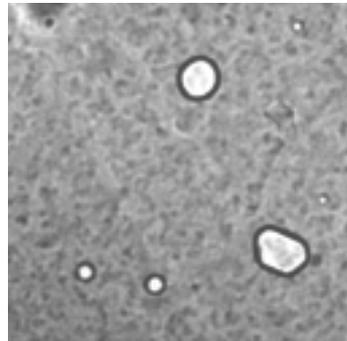


c) 800°C 4h55

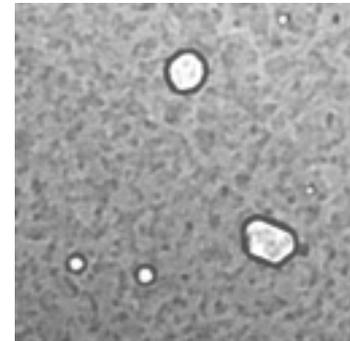
2nd Annealing



d) 800°C 4h50



e) 800°C 27h 10



f) RT ↘

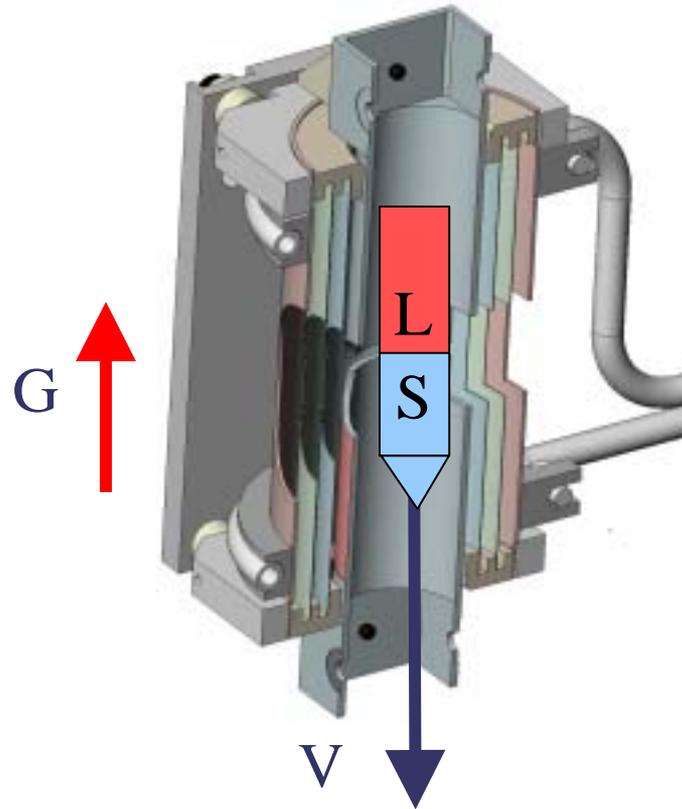
- Increase of the global perfection after annealing
- Most of pores reduce their sizes.
- Small pores are not anymore detectable
- Pores at surfaces did not shrink

results support the hypothesis of vacancy diffusion

Experimental device

Bridgman furnace with two independent heating elements

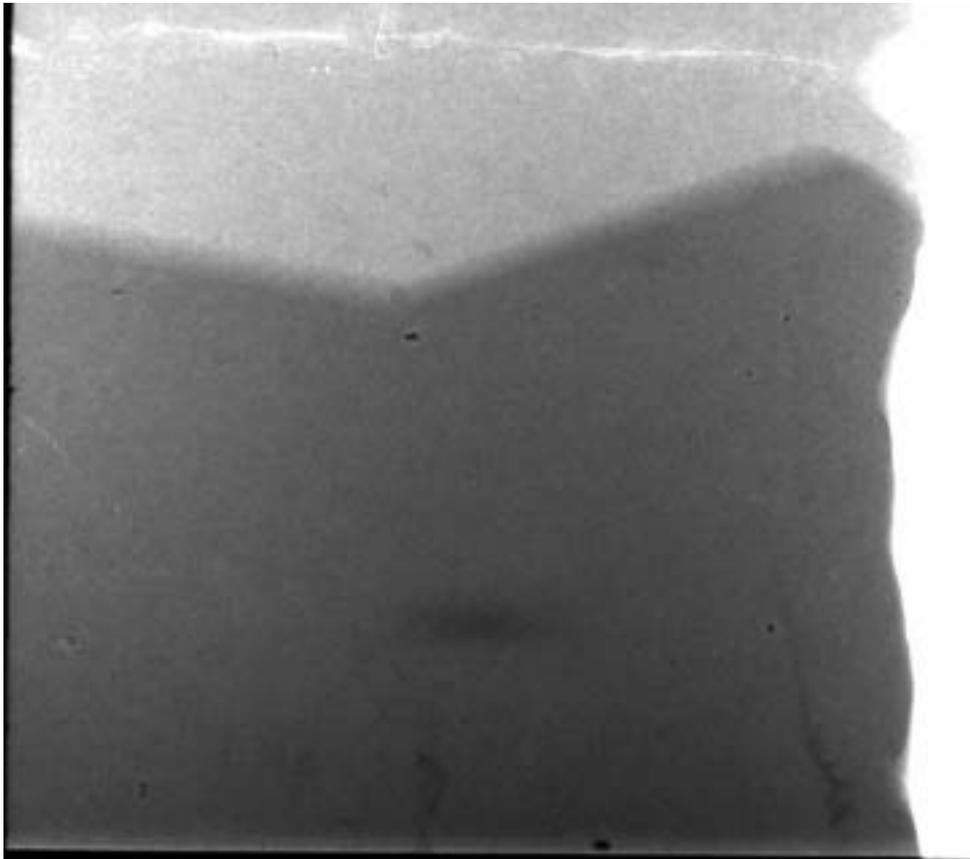
Sample size: 37 x 6.0 x 0.20 mm; Ultra vacuum ($10^{-7/-8}$ mbar)



Real-time observation of the growth of a AlPdMn quasicrystal

$\text{Al}_{68.8} \text{Pd}_{24.0} \text{Mn}_{7.2}$, $G = 20 \text{ K/cm}$, $V = 3.6 \mu\text{m/s}$

Real time length : 0h27

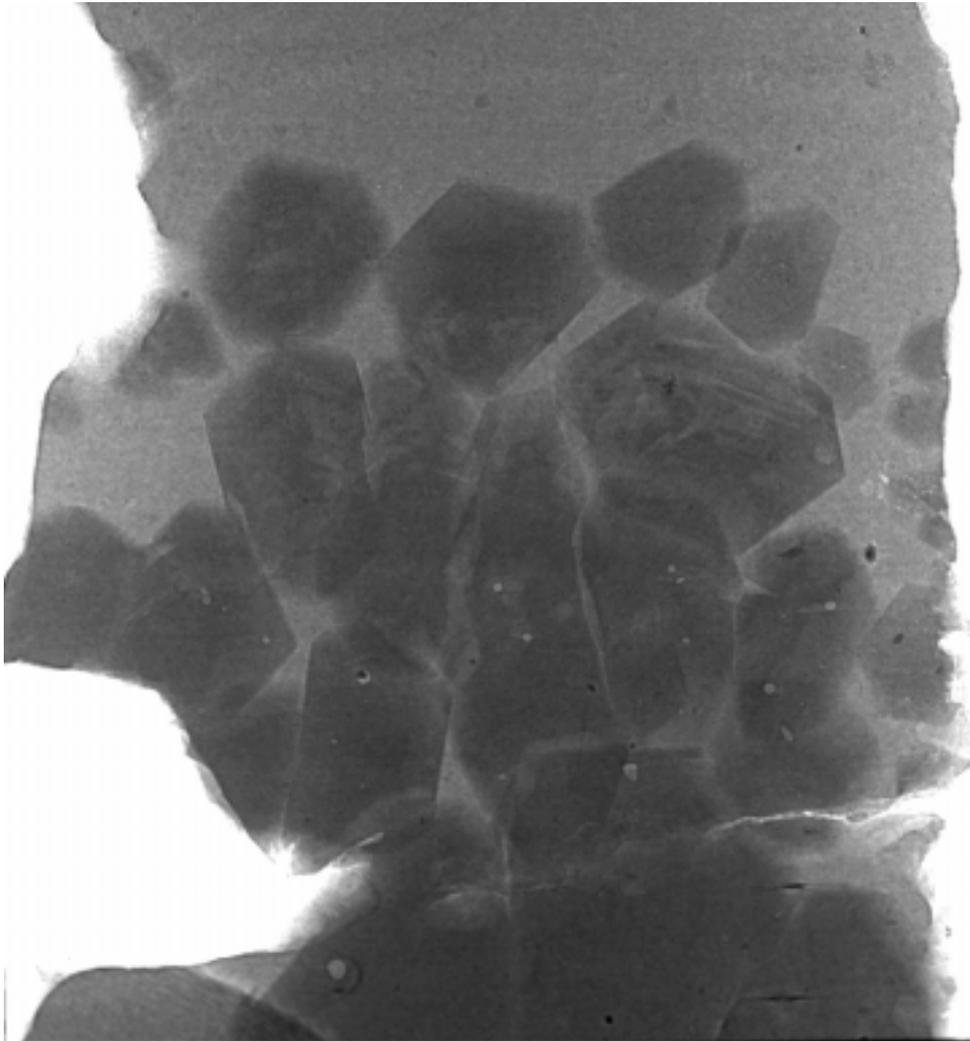


⇒ Nucleation of multiple grains ahead of SL interface

⇒ Step-like marks in each grain:
macrosteps, segregations?

H. Nguyen Thi, T. Schenk,
J. Gastaldi, J. Härtwig, H.
Klein, J. Baruchel (J. Cryst.
Growth, in press)

Real time observation of the growth of a Al-Pd-Mn quasicrystal



H. Nguyen Thi,
J. Gastaldi,
T. Schenk,
H. Klein,
J. Härtwig
(2004)

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not only imaging, but therapy

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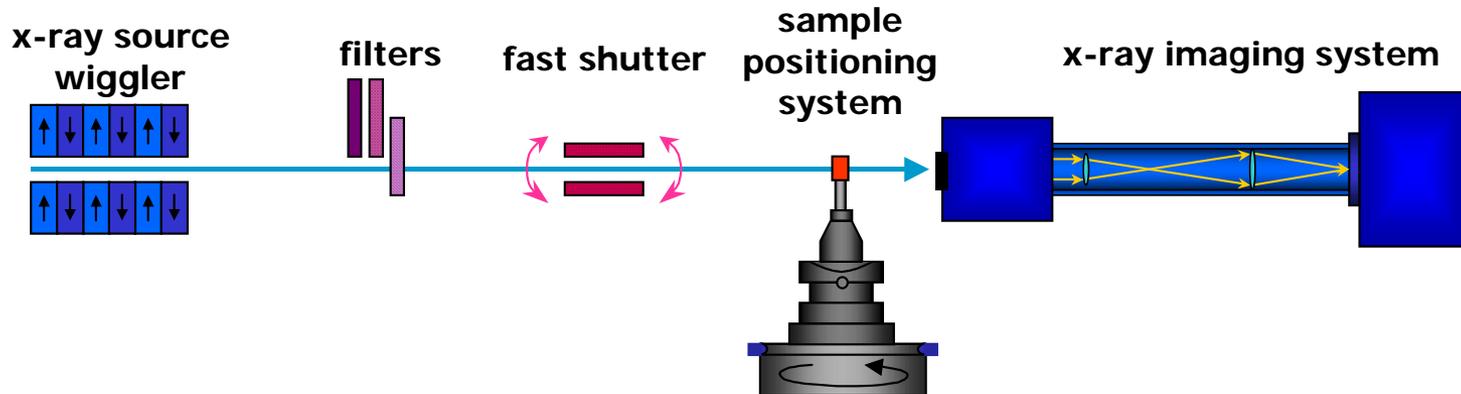
•3 – where should we go?

near future perspectives

FAST TOMOGRAPHY

ID15 Beam line at ESRF

Experimental device



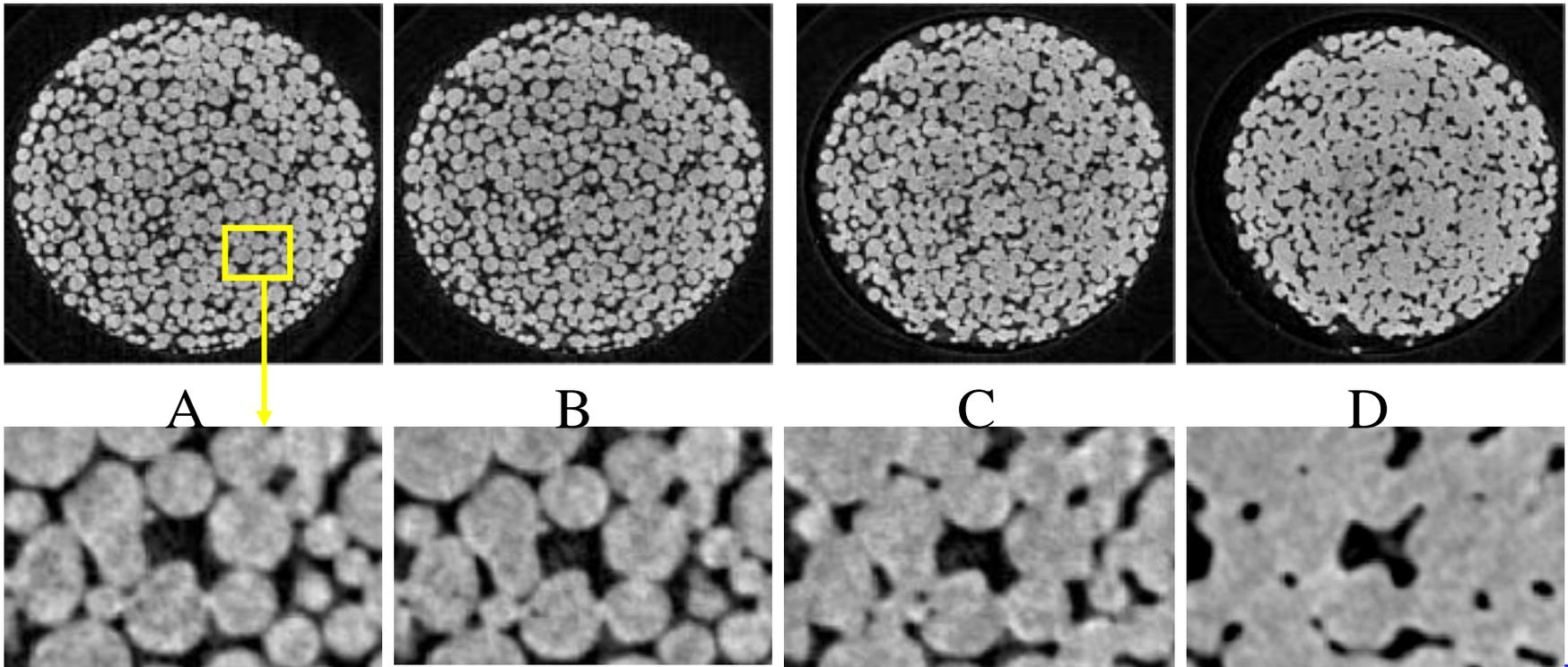
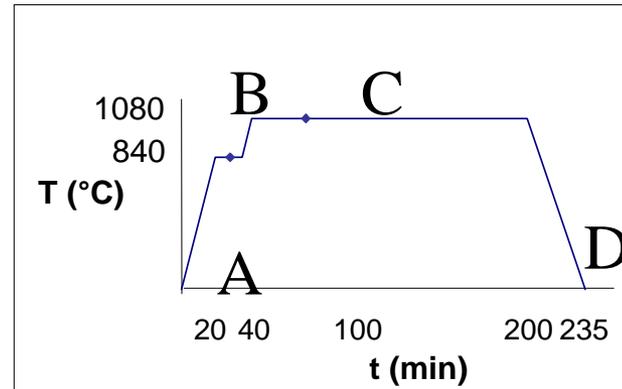
By adjusting the wiggler gap and choosing the proper filter the white beam spectrum is optimized in order to get the best signal-to-noise ratio

Exposure time per image: from 10 to 50 ms

In-situ follow of Cu powder sintering at 1050°C

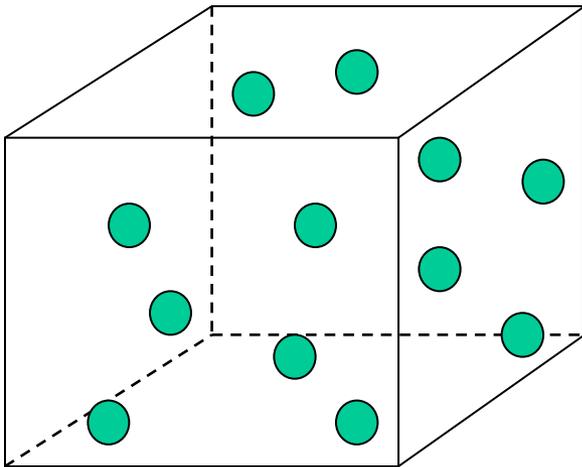
O. Lame, D. Bellet (INPG, Grenoble),
M. di Michiel (ESRF)

300 μm

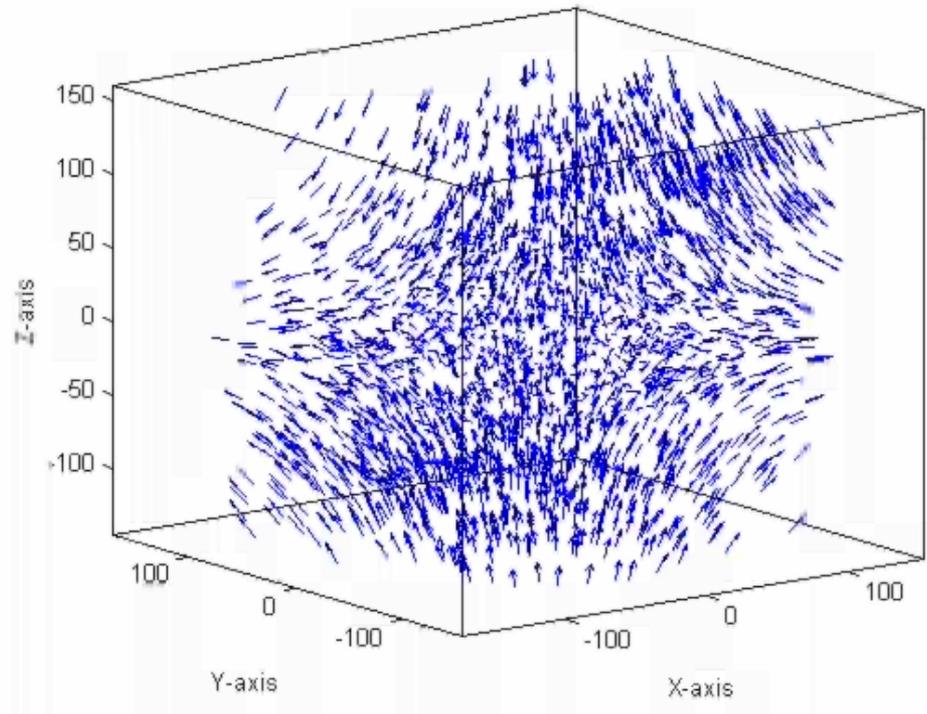


Tomography can reveal plastic strain field

Markers:



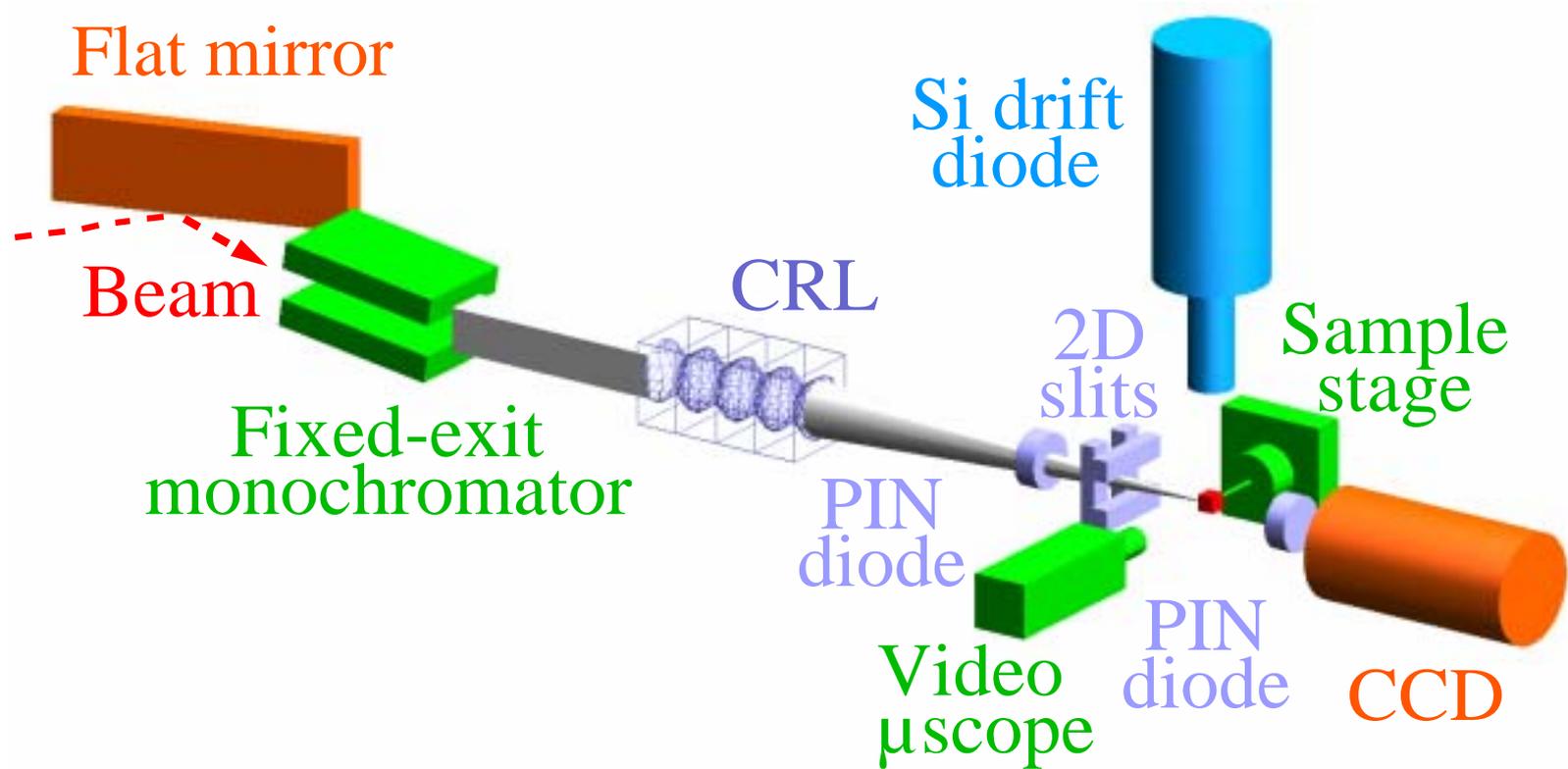
Compression of Al/W sample:



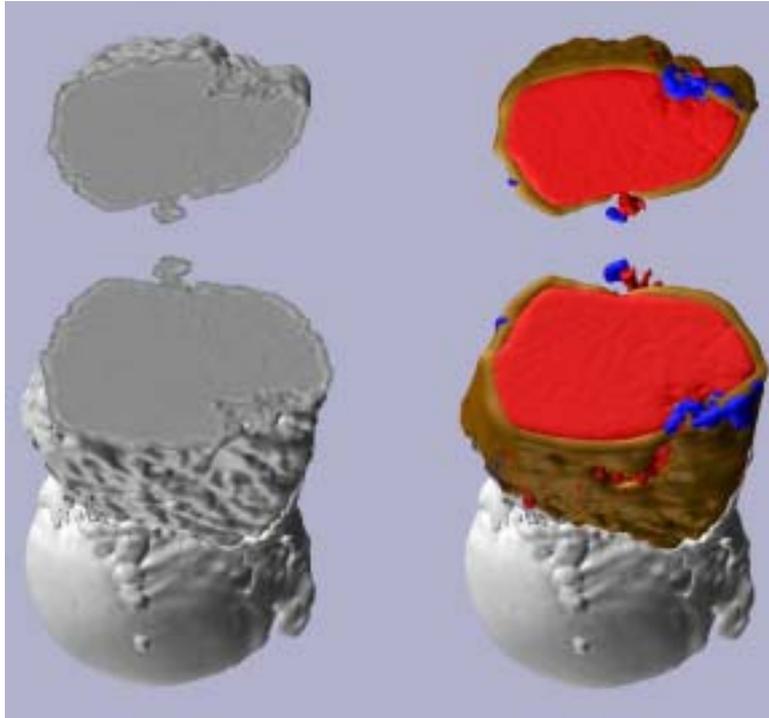
Work performed at HASYLAB (Germany)

S.F. Nielsen, F. Beckmann *et al.* *Acta Mater.* **51**, 2407 (2003)

Fluorescence tomography: towards 3D imaging



Combination with fluorotomography



3D rendering, fly ash particle

transmission tomography

**distribution of Rubidium (red),
Manganese (brown) and Iron
(blue)**

Voxel size $3 \times 3 \times 3 \mu\text{m}^3$.

B. Golosio, et al., *J. Appl. Phys.* **94**, 145-156 (2003) and *Appl. Physics Letters*, in press,

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Scientific Case

Nano-technology/fabrication: 50 nm scale and below

Materials Science: relevant scale 50 nm - 50 μm

Bio-materials: implants, ...

Soft condensed matter: soils, ...

Micro-structure, composition, plasticity

3D imaging, with improved temporal and spatial resolution

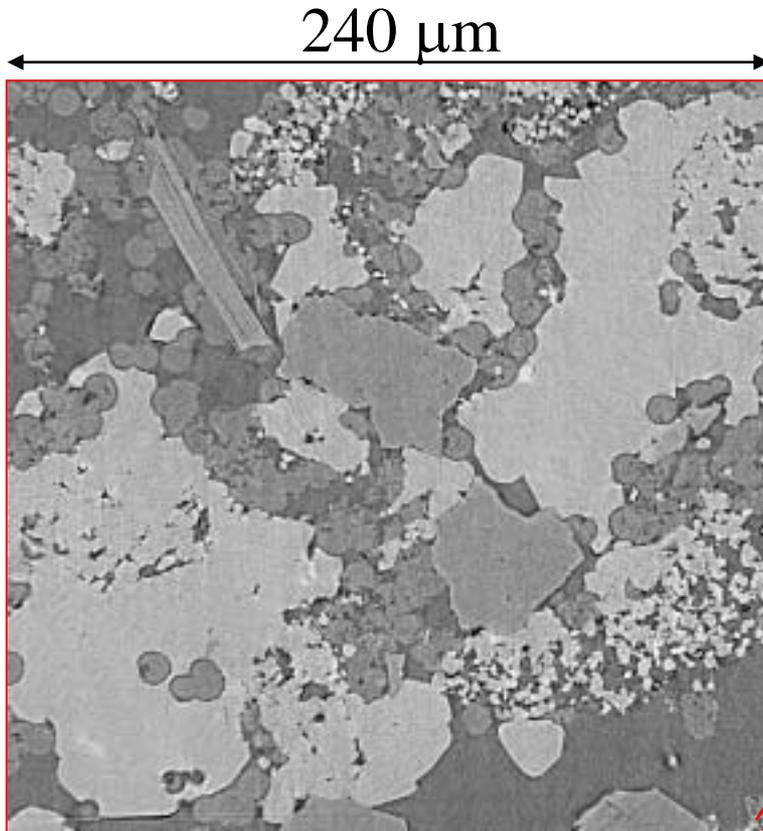
In-situ experiments with sample environment

Micro-analysis possibilities

The resolution gap (1)

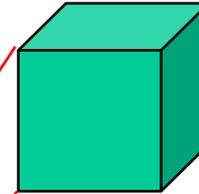
X-ray micro-tomography

O Rozenbaum, Univ. d'Orléans / CNRS-ISTO



1 voxel = 280 nm

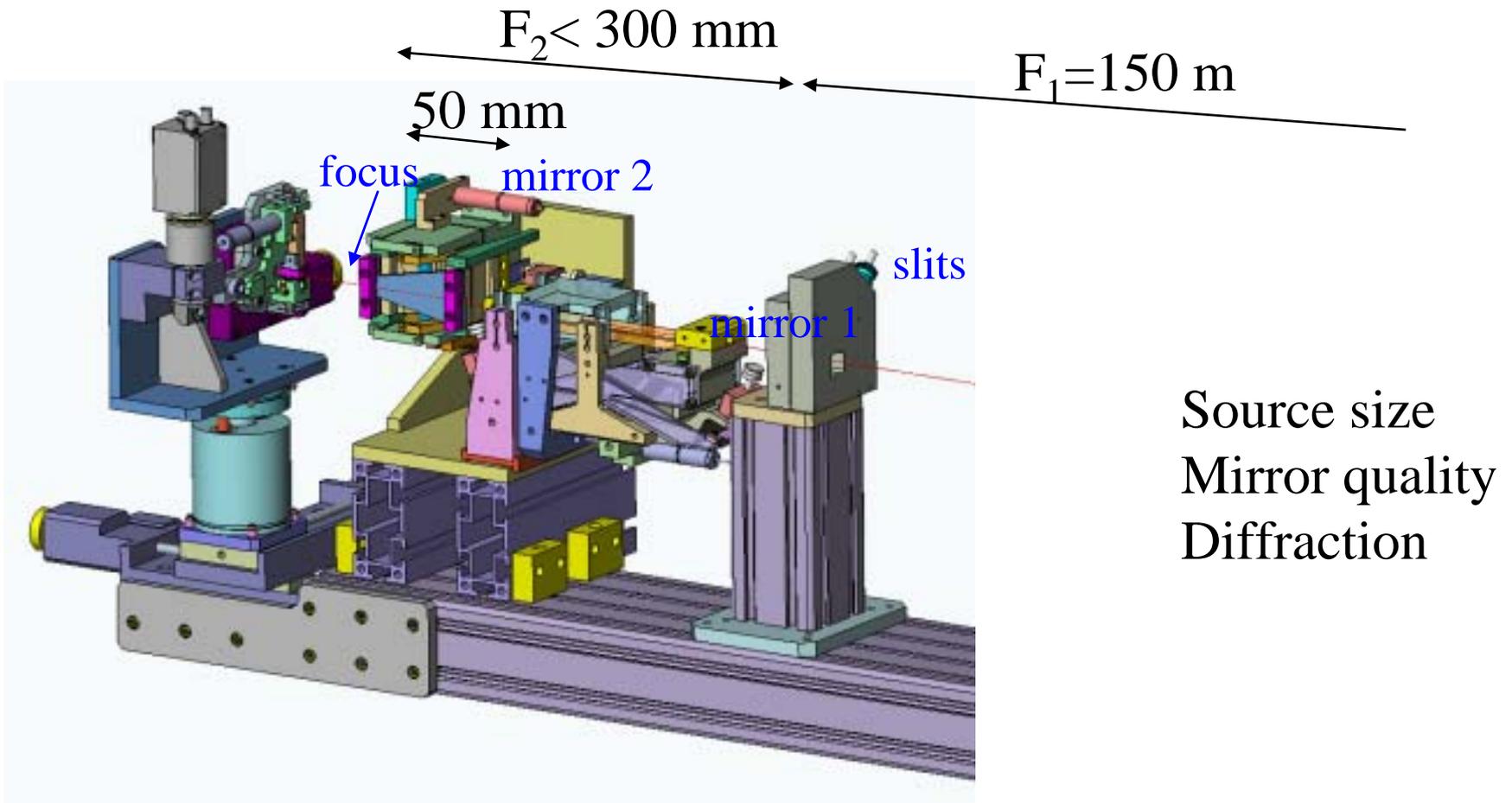
Tomography in STEM with 1 nm diameter electron probe.



Resolution about 1nm.

100 keV. P.Midgely et al, 2002

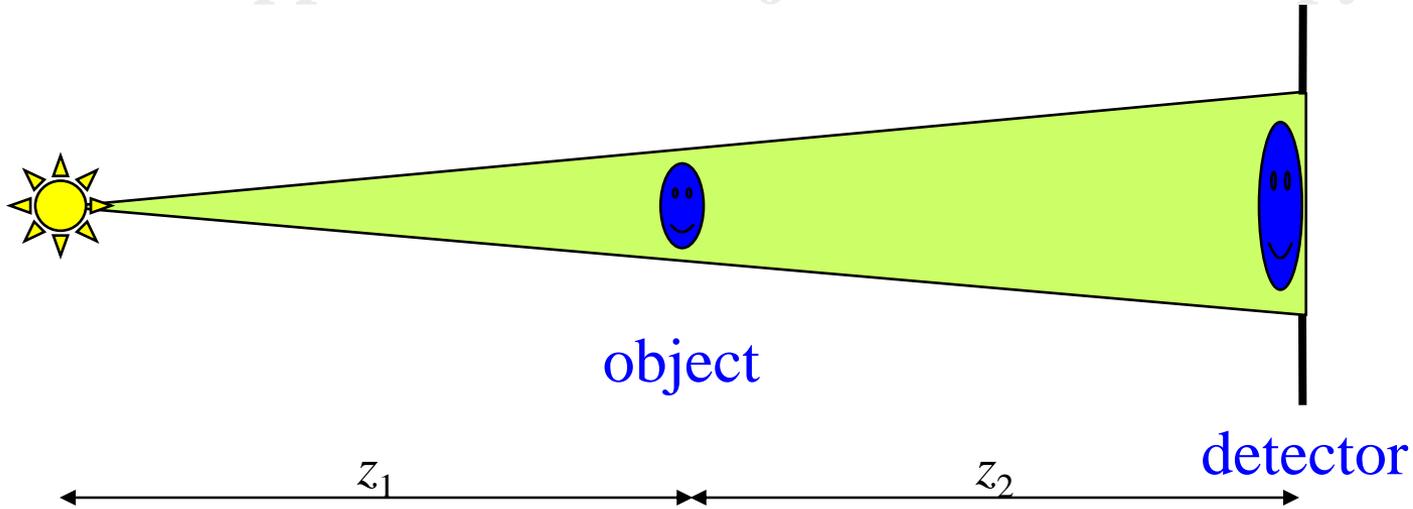
Kirkpatrick-Baez focusing (O. Hignette, P. Cloetens)



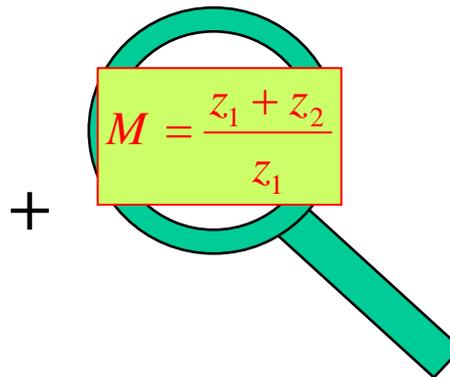
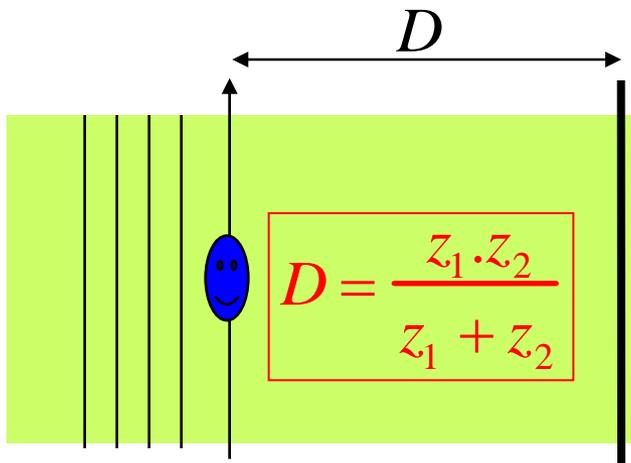
focus = demagnified image of the source; demagnification = F_1/F_2

Source size: $125 \mu\text{m}(\text{h}) * 25 \mu\text{m}(\text{v})$; **experimental result 70 nm x 90 nm**

Applications: Projection Microscopy



equivalent to



Limit cases

$$z_1 \gg z_2$$

$$D = z_2 ; M = 1$$

$$z_1 \ll z_2$$

$$D = z_1 ; M = z_2 / z_1$$

Plane wave illumination

Magnification

Projection Microscopy

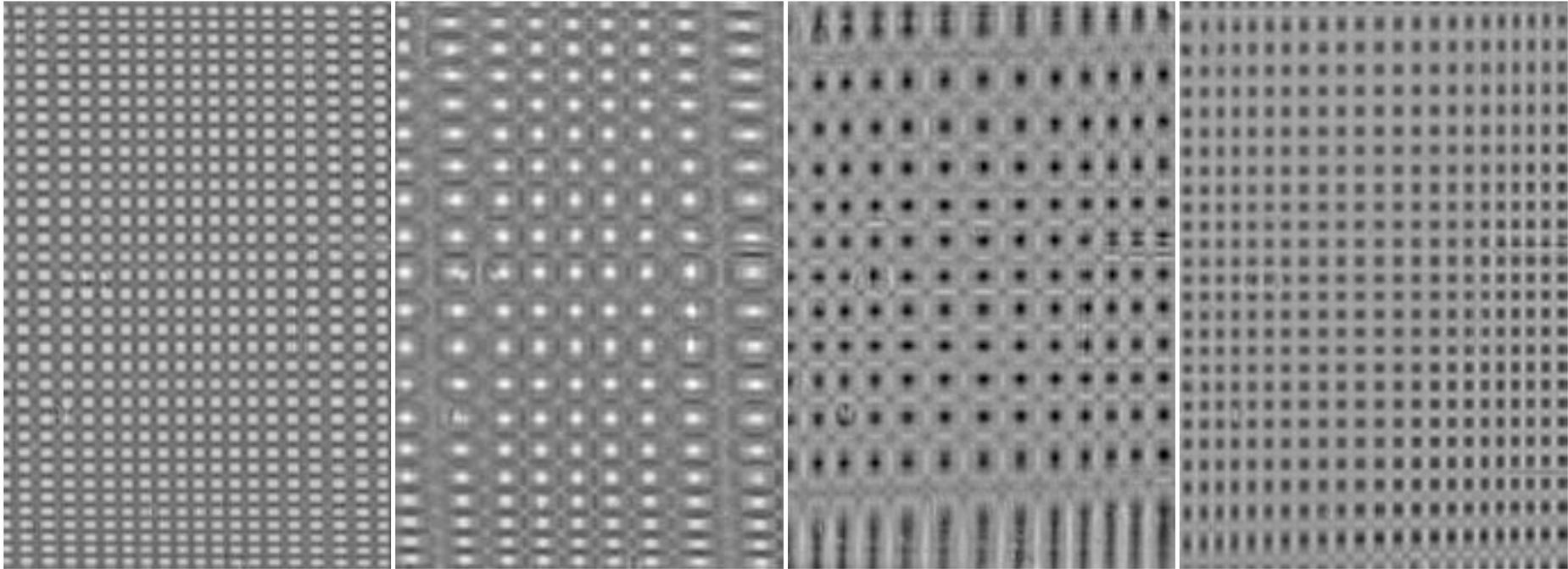
2 μm pitch grating (Si)

C. David, PSI

10 μm

Mirrors fully illuminated

0.2 (H) x 1.1 (V) mm, $E=20.5$ keV



$D = -20$ mm

$M = 115$

$D = -10$ mm

$M = 230$

$D = 10$ mm

$M = 230$

$D = 20$ mm

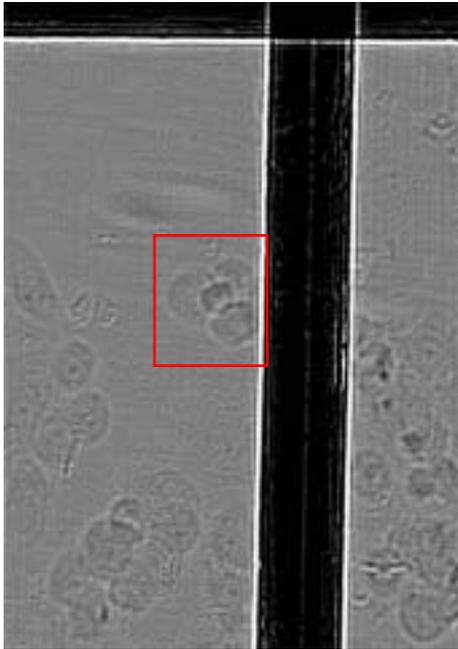
$M = 115$

focus

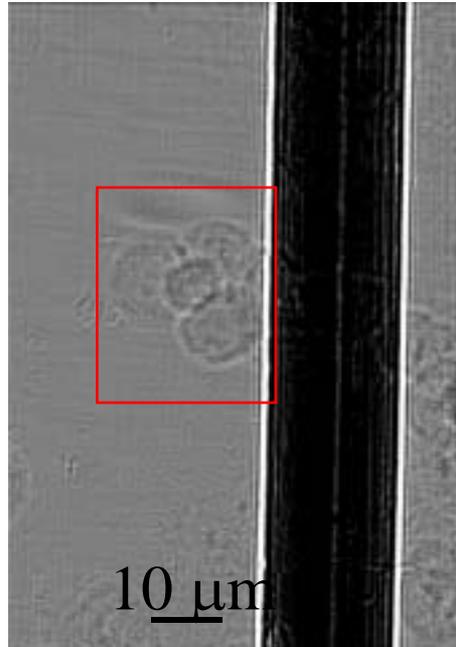
Exposure time = 0.4 s ! (16-bunch, saturation CCD)

P. Cloetens, O. Hignette

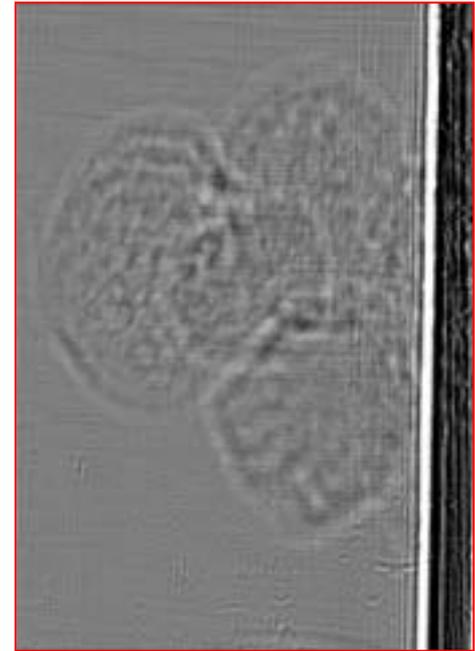
Projection Microscopy: cancerous cells



$D = 50 \text{ mm}$
 $M = 45$



$D = 30 \text{ mm}$
 $M = 75$



$D = 10 \text{ mm}$
 $M = 230$

$E = 20.5 \text{ keV}$; exposure time = 0.3 s ! (16-bunch)

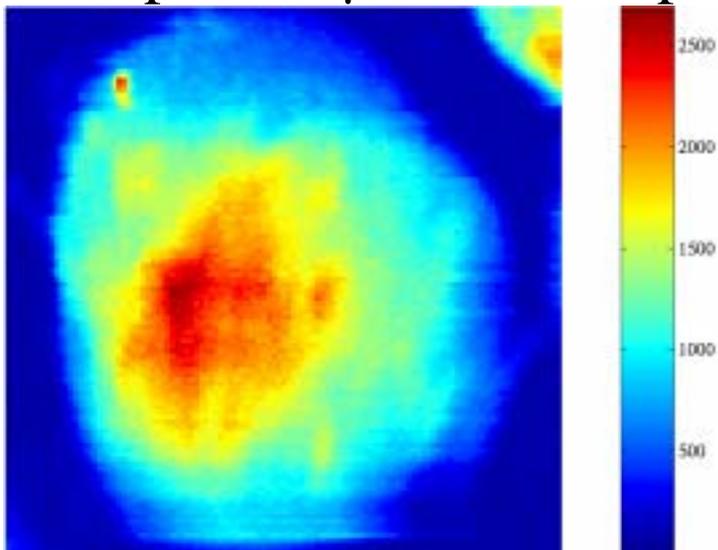
Problems: aberrations, instabilities, refraction by the sample itself,

Applications: Fluorescence

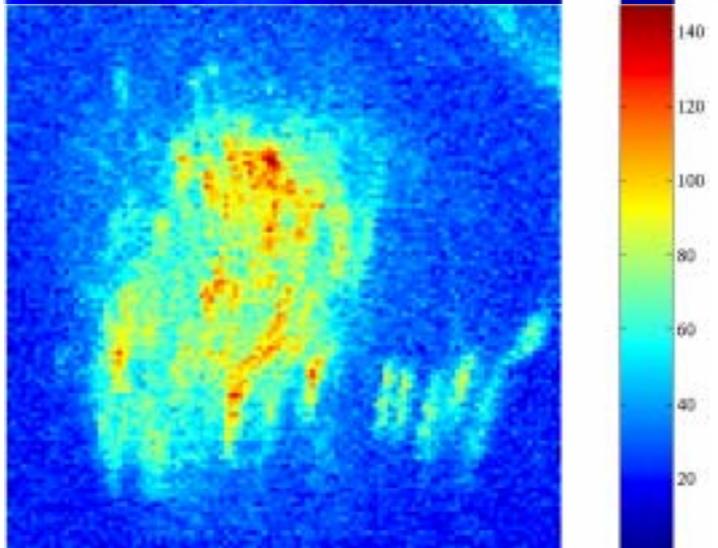
Cancerous cells

Step = $0.3 \mu\text{m}$, 0.9 sec/pt

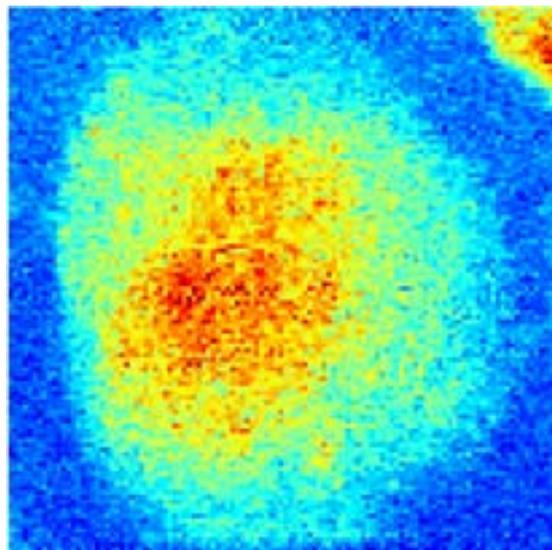
K



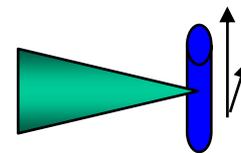
Fe



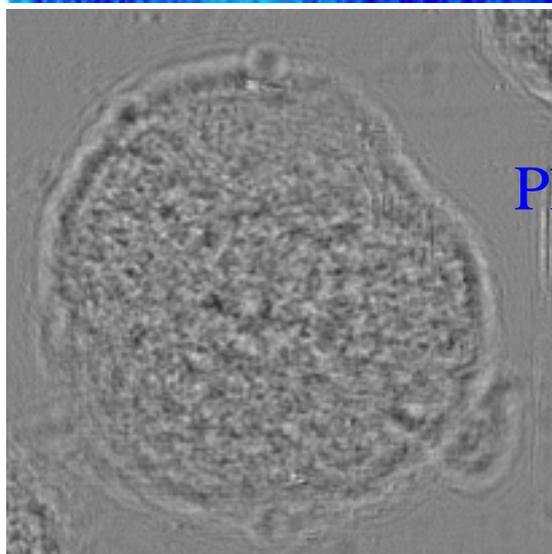
$40 \mu\text{m}$



Pt



Phase Contrast



P. Cloetens,
W. Ludwig,
S. Bohic

The resolution gap (2)

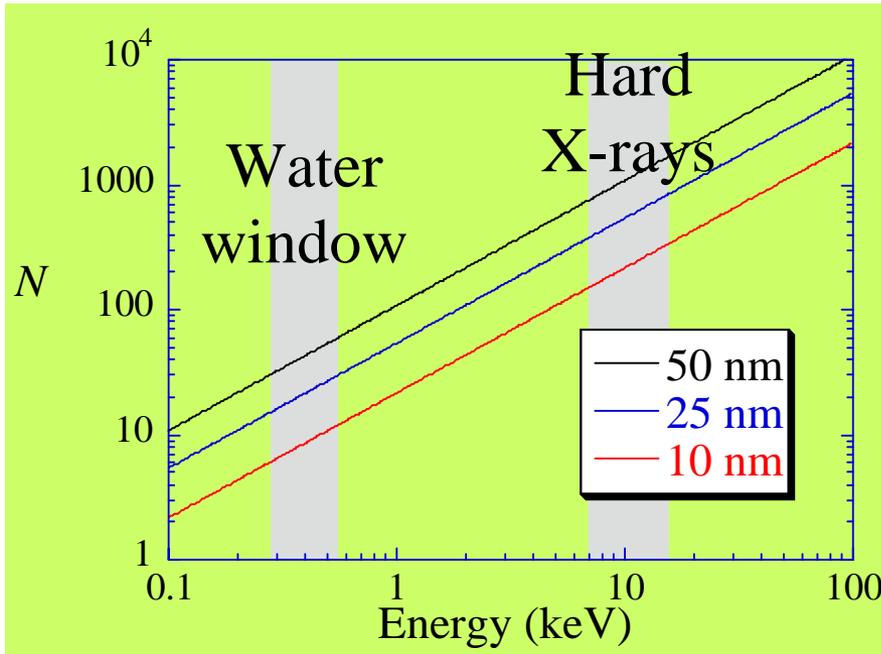
Can X-ray imaging become a true 3D ultra-microscopy technique?

$$N = \frac{\text{max. sample thickness}}{\text{resolution element}}$$

Absorption
Depth of focus
Multiple scattering

Optics
Interaction strength
Photon statistics

Depth of focus



50 nm resolution

water window ► $N = 60$
(= $3\mu\text{m}$)

10 keV ► $N = 1000$
(= $50\mu\text{m}$)

⇒ Soft X-ray microscopy remains 2D technique

⇒ 'Hard' X-rays will have to fill the gap



“nanotomography project” (Peter CLOETENS)

Conclusions

« X-ray Imaging » : series of (rapidly evolving) techniques that apply to a large variety of topics

The coherence of the beam adds new possibilities

We must improve the spatial and time resolution to be useful for many of the problems encountered in modern science and technology

Images contain a lot of information: the scientific use this information implies an accurate knowledge of the contrast mechanisms, and (often) image processing